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Effects of Active Gases on Droplet Transfer and Weld Morphology in Pulsed-Current NG-GMAW of Mild Steel

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Abstract

The current research of narrow-gap gas metal arc welding (NG-GMAW) primarily focuses on improving the sidewall fusion and avoiding the lack-of-fusion defect. However, the high cost and operation difficulty of the methods limit the industrial application. In this study, small amount of active gases CO₂ and O₂ were added into pure argon inert shielding gas to improve the weld formation of pulsed-current narrow-gap gas metal arc welding (NG-GMAW) of mild steel. Their effects on droplet transfer and arc behavior were investigated. A high-speed visual sensing system was utilized to observe the metal transfer process and arc morphology. When the proportion of CO₂, being added into the pure argon shielding gas, changes from 5% to 25%, the metal transfer mode changes from pulsed spray streaming transfer to pulsed projected spray transfer, while it remains the pulsed spray streaming transfer when 2% to 10% O₂ is added. Both CO₂ and O₂ are favorable to stabilizing arc and welding process. O₂ is even more effective than CO₂. However, O₂ is more likely to cause slags on the weld surface, while CO₂ can improve the weld appearance in some sense. The weld surface concavity in NG-GMAW is greatly influenced by the addition of active gas, but the weld width and weld penetration almost keep constant. This study proposes a new method which is beneficial to improving the weld bead formation and welding process stability.

Keywords: Metal transfer, Active gases, Pulsed-current NG-GMAW, Weld morphology

1 Introduction

Thick and large structural components are widely applied in various fields, such as offshore drilling platforms, shipbuilding, high-pressure vessels, and nuclear industry. Narrow-gap gas metal arc welding (NG-GMAW) is a preferential technology in manufacturing such components owing to its great advantages, such as less filler metal, higher productivity, low heat input and minimal welding distortion [1, 2]. However, the lack-of-fusion is the most common defect in NG-GMAW joints [3]. To improve the weld formation of NG-GMAW, plentiful progressive approaches have been put forward. He et al.

adopted a swing arc method to the narrow-gap welding and reported that the swing arc had a great influence on the weld formation and sidewall fusion [4]. Increasing the arc swing frequency and amplitude contributed to increasing the surface concavity and sidewall penetration. In addition, Zhu et al. showed that the high arc-swing frequency and amplitude made more heat transferred to the sidewalls, beneficial to achieving a more symmetric weld bead and reducing the pores in the bead [5]. Duan et al. carried out the rotating arc NG-GMAW by a hollow draft motor and curved contact tube [6]. They indicated that too large sidewall stay time could result in the lack-of-fusion defect and large rotation angular velocity could avoid the overlap defect. Wang et al. developed an alternating magnetic field to assist the arc welding in narrow groove, and successfully obtained a larger sidewall penetration with a magnetic flux density up to 6 mT [7].

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Besides, laser-arc hybrid welding and tandem GMAW are also commonly utilized in narrow gap welding processed to eliminate the lack-of-fusion defect and improve the weld formation [8–11].

All of these methods mentioned above can improve the weld bead formation but need specially designed welding torches or additional equipment, which means more investment. It is of great significance to exploit a cost-effective method to improve the weld formation of NG-GMAW. As is known, shielding gas is a significant factor influencing the GMAW process. Argon is the most commonly used shielding gas in arc welding, since it has no reaction with the molten metal. However, it may cause arc instability and low wettability [12]. To achieve better joint performances, varieties of mixtures with inert and active gases were developed. Zhang et al. studied the influence of shielding gases on the weld shape of a hybrid welding [13]. It was reported that the weld bead formation was improved first and then worsened and the weld penetration increased with the CO₂ content in pure Ar increasing from 0% to 100%. Huang reported that adding nitrogen into an argon-based shielding gas would increase the heat input [14]. Both the weld penetration and weld area increased with increase of nitrogen in argon-based shielding gas. Moreover, it was reported that the pores in the laser welds could be eliminated by using N₂ as the shielding gases [15]. Zong et al. studied the influence of shielding gas on undercut defects in GMAW and found that the undercut defects could be significantly reduced by increasing the CO₂ content [16]. However, when the CO₂ content was higher than 20%, welding spatters occurred frequently on the weld bead surface. The variation of weld bead formation is attributed to that the shielding gas directly determines the arc behavior and surface tension which influence the metal transfer and fluid flow in the welding pool [17–19]. Cai et al. evaluated the effect of ternary mixtures of Ar + He + CO₂ on metal transfer in narrow-gap tandem GMAW. The results revealed that the metal transfer always remained a pulsed spray transfer when CO₂ was kept at 10%, no matter how the helium content varied in a certain range [20]. Zhao et al. found that decreasing the CO₂ content in Ar + CO₂ + O₂ led to a wider arc and a stable pulsed streaming spray in pulsed GMAW [21]. Similar results were also reported by Wong et al. [22].

The current research effort of NG-GMAW mainly focuses on improving the sidewall fusion and avoiding the lack-of-fusion defect. However, the methods proposed are not widely available due to the high cost for developing a reliable suitable narrow-gap GMAW torch and its related equipment. A proper selection of shielding gases can be an economical effective way to solve this problem, however the related studies are less reported in

NG-GMAW. In this research, a series of welding experiments were carried out to evaluate the effects of varied gas compositions on the arc behaviors, metal transfer, welding process stability and weld morphology in pulsed current NG-GMAW of ferrous metal.

2 Experimental Procedure

Figure 1(a) briefly shows the schematic diagram of welding experiment system setup, in which a new-designed narrow-gap GMAW torch was applied. A high-speed video camera was used to observe arc profile and droplet transfer process. Its frame rate was 3000 fps and the exposure time was 1.0 μ s. The viewing direction was parallel to the welding direction and the distance from the camera lens to the filler wire tip was about 0.5 m. An optical filter was used to block the arc light while capturing the droplet transfer process. The waveforms of welding current and arc voltage were detected by hall sensors and were sampled by a data acquisition system (DAS). Its sampling rate was 20 kHz. The welding experiments were conducted on the specimens of commercial quality hot-rolled steel Q345d, and ER50-6 wire with a diameter of 1.2 mm was used as filler metal. Their chemical compositions are given in Table 1. A ‘simulated’ narrow-gap workpiece was used with the dimension of 100 mm (length) \times 40 mm (width) \times 35 mm (height) with a groove as shown in Figure 1(b). It was simple but suitable for observing the metal transfer and arc behaviors. A power source working in direct-current electrode positive (DCEP) was applied to welding process, which was set in single pulse mode, as shown in Figure 1(c). Different kinds of shielding gases were adopted, which are mainly the mixtures of argon and CO₂ or O₂ respectively. The CO₂ content varied from 5% to 25% in the gas mixtures of Ar + CO₂, while the O₂ content changed from 2% to 10% in the gas mixtures of Ar + O₂. The gas flow-rates through the external shielding gas hood and the nozzle of welding torch were both 30 L/min with the total gas flow rate of 60 L/min. The welding current was 350 A, the arc voltage was 34 V, and the travelling speed was 300 mm/min. The wire extension was about 15 mm.

For investigation of welding process in a deep and narrow groove with varied gas mixtures, only the first pass of weld bead was deposited on each specimen. After welding, all the specimens were cut along the transverse direction of welding bead. Each specimen was polished and then was etched with 4% Nital solutions (4% nitric acid in alcohol). The weld bead was characterized by weld width (W), weld penetration (H) and weld surface concavity (C) for evaluating the influence of varied gas mixtures on the weld morphology, as shown in Figure 1(d). In addition, the arc shape was also characterized by arc width and arc length.

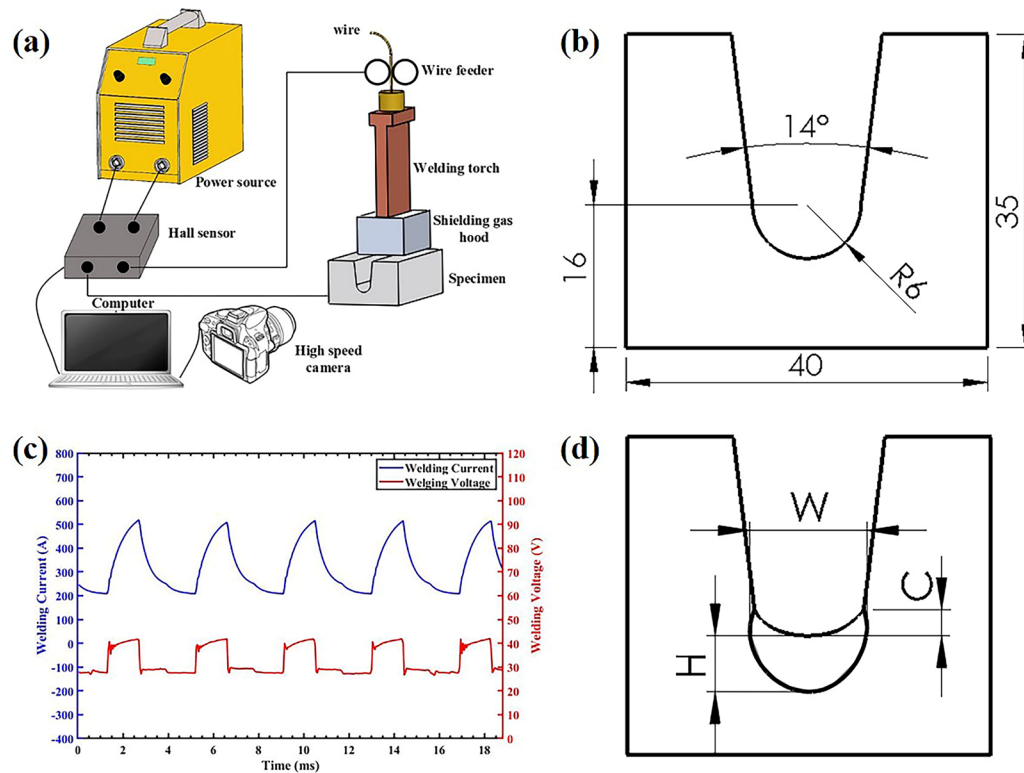


Figure 1 **a** Schematic diagram of welding experiment system setup, **b** cross section dimension of specimen and its narrow-gap groove, **c** Welding current and arc voltage waveforms, **d** characterization for first-pass weld bead of NG-GMAW

Table 1 Nominal chemical compositions of base metal and filler metal (wt %)

Material	Mn	Si	Ni	Cr	Ti	Al	V	S	P	Mo	Cu	C	Fe
Q345D	1.45	0.27	0.07	0.05	0.05	0.02	0.01	0.01	—	—	—	≤ 0.18	Bal.
ER50-6	1.40–1.85	0.80–1.15	≤ 0.15	≤ 0.15	—	—	≤ 0.15	≤ 0.025	≤ 0.025	≤ 0.15	≤ 0.5	0.06–0.15	Bal.

3 Results and Discussion

3.1 Arc Shape and Metal Transfer

Figure 2 presents the variation of arc profile under shielding of gas mixtures with varied CO_2 content. It indicates that the arc shape experiences large changes with increased CO_2 content. When the CO_2 content in the shielding gas is 5%, the arc extends even to the sidewalls of groove, which signifies that more heat is transmitted to the sidewalls and the welding pool is enlarged. The arc experiences a fluctuation in the range of welding current from base to peak. The fluctuation of arc is consistent with the current variation. The arc is cone-shaped at the base current, and bell-shaped with the arc edge upward warped at the peak current. The upward warp of arc mainly results from the restrain effect of narrow gap groove. The arc profiles under

shielding gas of $\text{Ar} + 5\% \text{CO}_2$ and $\text{Ar} + 10\% \text{CO}_2$ are correspondingly similar at both base and peak current. However, when the CO_2 content in the shielding gas is 25%, the arc keeps the same conical shape through the whole pulse cycle, except for the change of its size.

The effect of O_2 on the arc profile is shown in Figure 3. When the O_2 content in the shielding gas is 2%, the arc profile changes greatly comparing to that of any other gas mixtures mentioned above. The arc looks much like cone shape at the peak current, but slender bell-like shape at the base current. When O_2 is added up to 5% and 10%, the arc turns to be conical shape at the base current, and bell shape at the peak current, which is similar to that of gas mixtures with 5% or 10% CO_2 contents. As shown in Figures 2 and 3, in comparison with conventional GMAW, the sidewalls of narrow

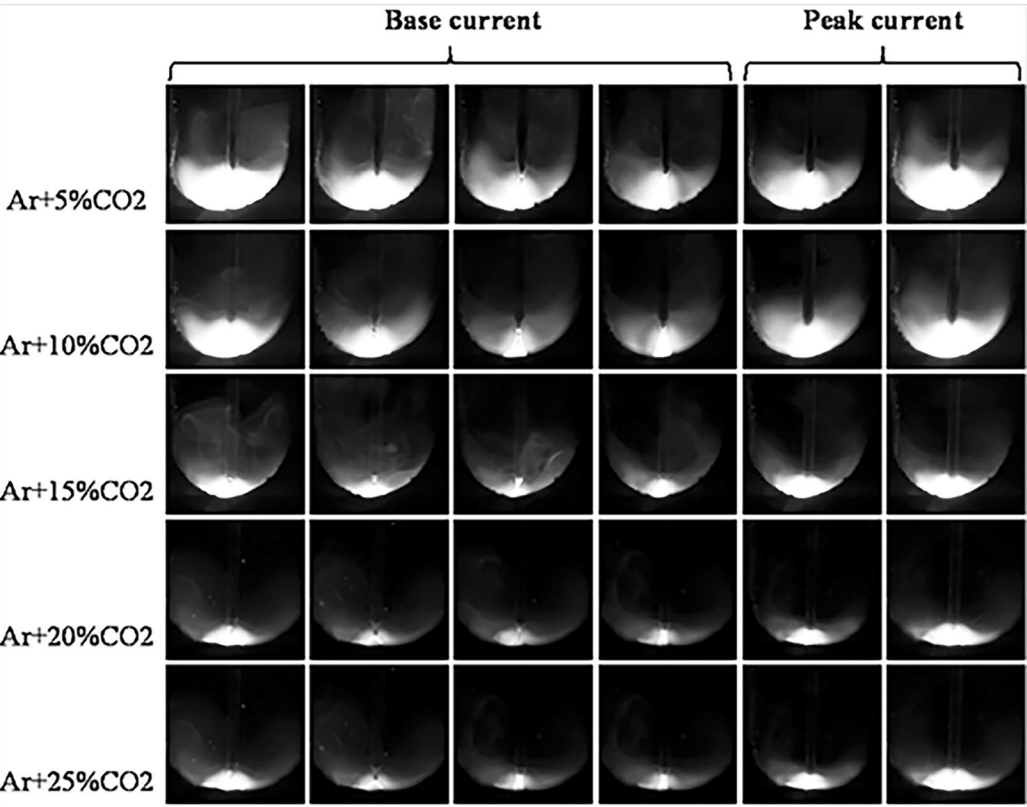


Figure 2 Variations of arc profile under shielding of gas mixtures with varied CO₂ contents

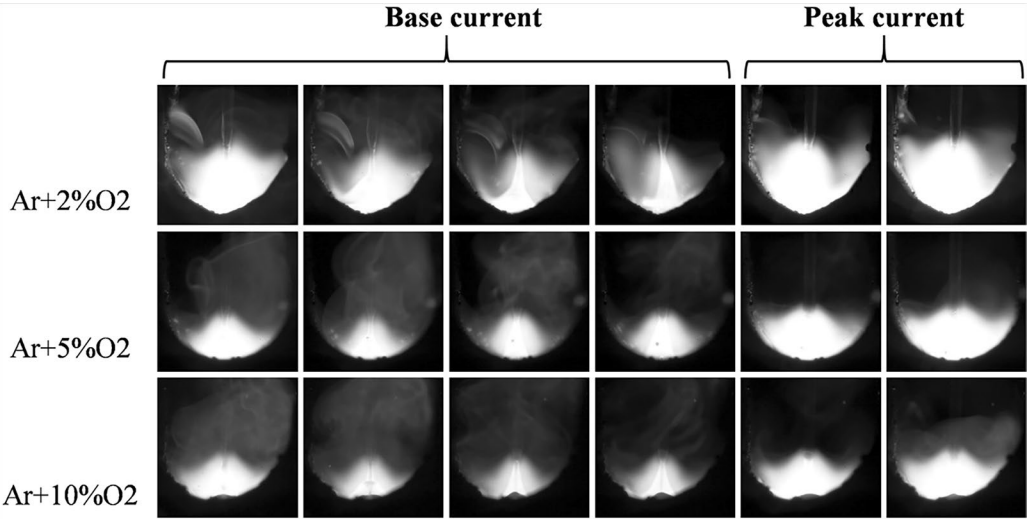


Figure 3 Variations of arc profile under shielding of gas mixtures with varied O₂ contents

gap groove have strong restraint on the welding arc. The arc could not extend freely, and has no choice but to climb up when touching the sidewalls.

The detailed arc profile characteristics such as arc length, arc width and arc area at the peak current for varied contents of CO₂ and O₂ in shielding gas are displayed in Figure 4(a) and (b) separately. An overall tendency is

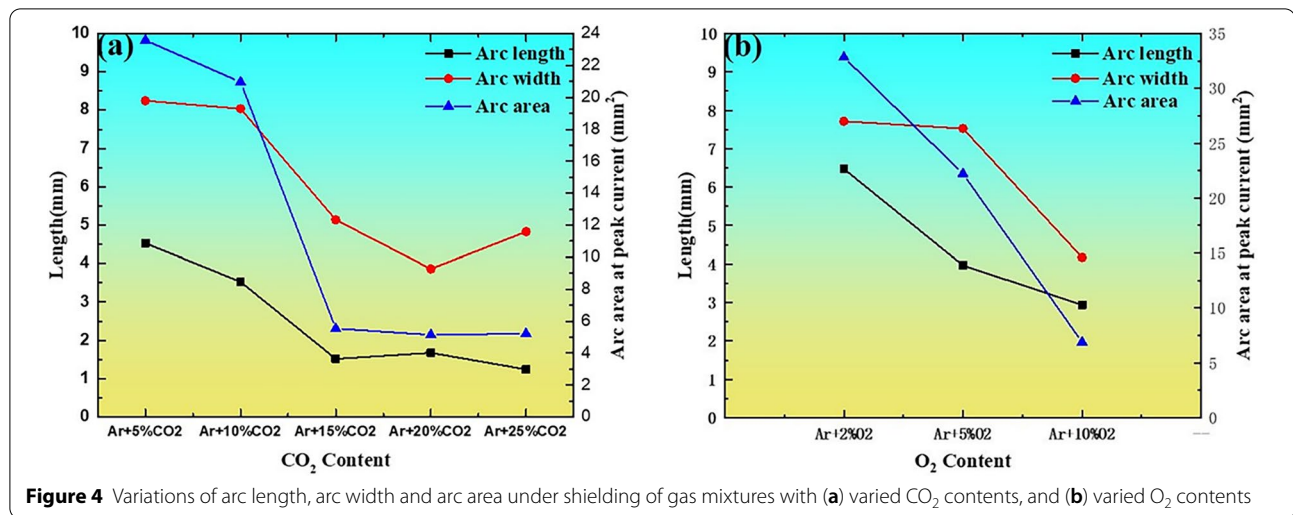


Table 2 Chemical and physical properties of gases used in pulsed current NG-GMAW

Shielding gas	Density at 15 °C 1 bar (kg/m ³)	Dissociation energy (eV/molecule)	First ionization energy (eV/molecule)	Chemical activity
Ar	1.669	0	15.8	Inert
CO ₂	1.849	4.3	14.4	Oxidizing
O ₂	1.337	5.1	13.6	Oxidizing

that all of them fall off with the increase of CO₂ or O₂. On the one hand, both CO₂ and O₂ are active gases, which is beneficial to stabilizing the cathode spot and obtaining a stable arc combustion when proper contents of them are added into the pure argon.

On the other hand, the temperature of arc edge is approximately 10000 K, where the thermal conductivity of O₂ is 1.21 W/m·K while the thermal conductivity of pure argon is only 0.66 W/m·K presented by Murphy et al. [23]. Wang et al. reported that the thermal conductivity of CO₂ is 1.47 W/m·K at the same temperature [14]. Therefore, when proper small amount of CO₂ or O₂ is added into the pure argon, the thermal conductivity of arc gets higher, which results in more uniform heat distribution and expansion of arc. However, much higher content of CO₂ also means more heat loss by conduction of arc because of its higher thermal conductivity. Meanwhile, CO₂ and O₂ are polyatomic molecules and need to be dissociated firstly before ionizing, as shown in Table 2, which means more energy needs to be consumed to ignite the arc. In addition, the decomposition and ionization of CO₂ will generate more ions to increase the arc electric field intensity, which results in increased electromagnetic force. When the CO₂ or O₂ content in

the shielding gas gets higher, the thermal pinch effect together with increased electromagnetic force constricts the arc, and the arc gets difficult to expand.

In light of above analysis, the shielding gas composition has a great impact on arc shape and its stability. Actually, it also plays a key role in metal transfer process of welding. When the shielding gas is changed to Ar + 5% CO₂ or Ar + 10% CO₂, the metal transfer has similar features at same pulsed-current. Figure 5 presents the pulse streaming spray transfer in welding with both shielding gas mixtures. The wire tip turns to be a tapered shape that represents a typical feature for streaming spray transfer. It is encircled by the arc, and is adequately heated to become plastic. As a result, a slender long molten neck hanging on the wire tip is formed and directs towards the weld pool. At the wire tip, a droplet with the diameter closing to the wire diameter is transferred from a cone-shaped wire tip into the welding pool in spray way. In addition, the arc jumping phenomenon along with the long neck is also observed in Figure 5, and the current at this moment is defined as the transition current of streaming spray transfer.

In pulsed streaming transfer process, it can be seen that a plenty of small spatters are produced in the shielding gas mixture of Ar + 5% CO₂ as shown in Figure 5(a). This is owing to the relatively long arc in Ar + 5%CO₂, which means lower arc stiffness. It brings about arc fluctuation, and thus the force balance of the radial component of the electromagnetic force is easy to be broken. One more to note is the arc jumping phenomenon which took place at $t + 2.67$ ms and $t + 4$ ms as shown in Figure 5(a). The arc root is firstly located at the bottom of neck as shown in Figure 6(a), and gradually climbs to the top of the neck which becomes longer and thinner as shown in Figure 6(b). At the same time, the arc also

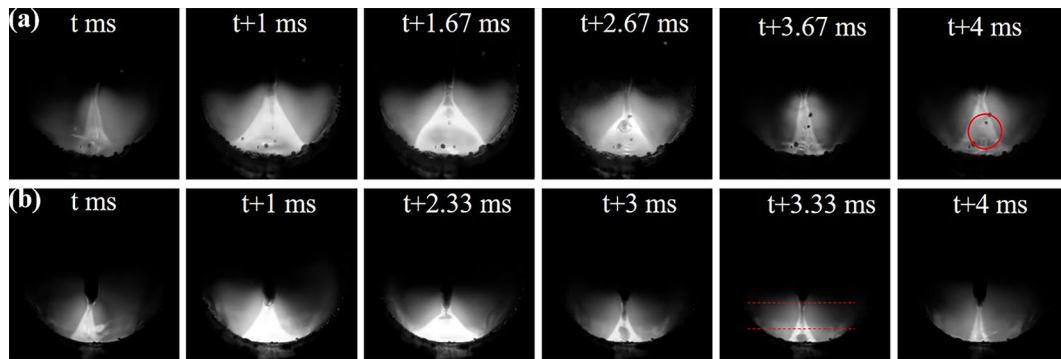


Figure 5 Metal transfer process under shielding of gas mixtures of (a) Ar + 5% CO₂ (b) Ar + 10% CO₂

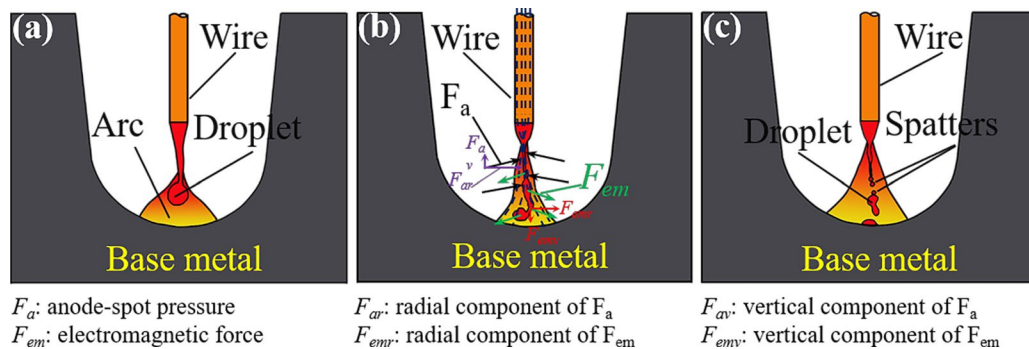


Figure 6 Schematic illustration of arc jumping phenomenon: **a** arc surrounding the bottom of droplet, **b** force analysis when arc climbs to the top of neck, **c** droplet transfer after arc jumping

changes to conical shape. The longer and thinner neck is surrounded and directly heated by the arc, and thus leads to a large anode-spot pressure acting on the thinner neck. Under the combined action of the anode-spot pressure and unbalanced electromagnetic force, the neck gets unstable. Under this condition, when the droplet is transferred to welding pool, the neck is broken at one point on the neck rather than on the wire tip. This causes the generation of small droplets, which ultimately form spatters rather than are transferred into the welding pool (as shown in Figure 6(c)).

With CO₂ increasing to 15%, it can be observed in Figure 7(a) that the neck is shorter than that in the shielding gas mixture of Ar + 5% CO₂ or Ar + 10% CO₂ as shown in Figure 5, the metal is transferred in projected spray-like transfer mode. When Ar + 20%CO₂ or Ar + 25%CO₂ is applied, the metal is transferred in projected spray transfer mode as shown in Figure 7(b) and (c), in which a droplet diameter is roughly equal to the wire tip diameter.

When the CO₂ content is relatively low (5% or 10%), the energy for arc ignition is just a little higher than that in the pure argon, and so is the heat loss because of its high thermal conductivity. However, when CO₂ content

increases up to 25%, the thermal conductivity gets great improvement, consequently, the heat loss increases sharply, and so does the energy for arc ignition. Moreover, the variation of arc behavior mentioned above also have a great impact. The arc just hangs on the droplet without climbing up to the wire. Consequently, the energy for heating the wire tip is not enough to form a streaming.

The droplet transfer mode is apparently influenced by different arc behaviors under varied shielding gases, and it is virtually influenced by the force distribution on the droplet. The force balance theory claims that a droplet will detach from the electrode tip when the detaching forces exceed the retention forces.

Generally, four different forces are considered in the metal transfer of arc welding. The surface tension is a retention force, while the gravity, the plasma drag force and the electromagnetic force are detaching forces. Nemchinsky [24] proposed that the drag force is not important during droplet transfer process at high currents. According to Scotti et al. [25], the electromagnetic force is the governing force during both projected spray and streaming spray transfer, and its induced pinch effect

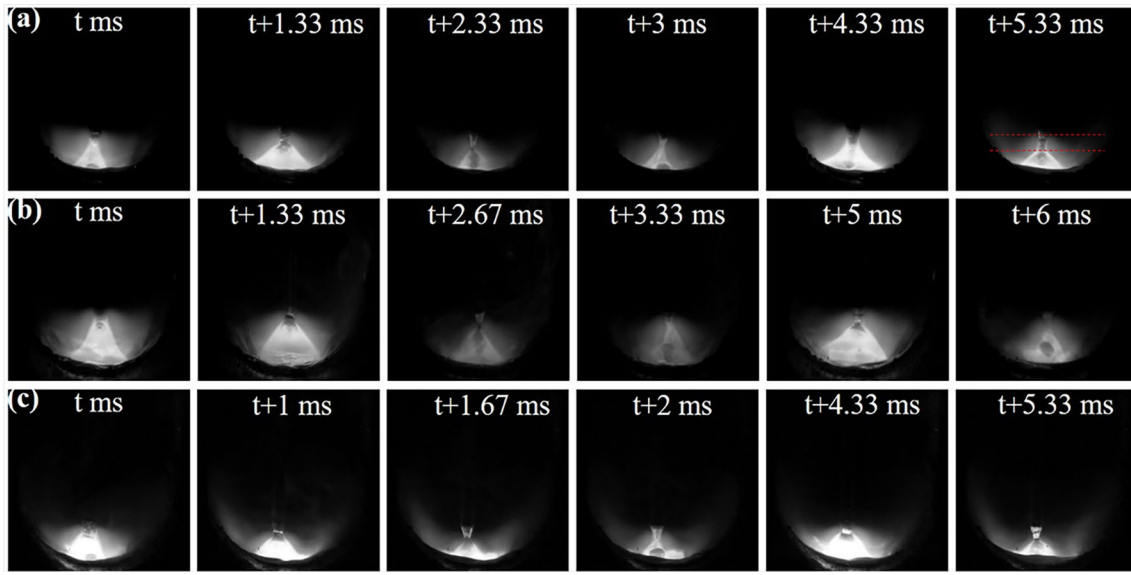


Figure 7 Metal transfer process under shielding of gas mixtures of (a) Ar + 15% CO₂, (b) Ar + 20% CO₂, (c) Ar + 25% CO₂

limits the droplet. So more attention is put on surface tension and electromagnetic force.

The gravity F_g in flat position welding is given by Kim and Eagar as below [26]:

$$F_g = \frac{4}{3}\pi r_d^2 \rho_d g, \quad (1)$$

where r_d is the droplet radius, ρ_d is the mass density of the droplet, and g is the gravitational acceleration. The surface tension can be expressed as follows:

$$F_\gamma = 2\pi R_e \gamma, \quad (2)$$

where R_e is the radius of the electrode and γ is the surface tension coefficient of the liquid metal, which is significantly influenced by the O₂ content in a binary Fe-O system according to Sahoo et al. [27].

The plasma drag force on the droplet is given by [28]:

$$F_d = C_d \frac{1}{2} \rho_g v_g^2 \pi r_d^2, \quad (3)$$

where C_d is the drag coefficient, ρ_g is the plasma gas density, v_g is the gas velocity and r_d is the radius of the droplet.

The vertical component of electromagnetic force suitable for the droplet transfer is given by Amson as follows [29]:

$$F_{env} = \frac{\mu_0 I^2}{4\pi} \left(\frac{r_d}{r_w} \right)^2 f, \quad (4)$$

$$f = \frac{1}{4} - \left(\frac{3}{2} - \ln 4 \right) \left(1 + \frac{l_d}{r_d} \right)^{-2}, \quad (5)$$

where μ_0 is the permeability in free space, I is the welding current, r_d is the radius of the droplet, r_w is the radius of the wire and l_d is the length of the radius.

Figure 8 illustrates the arc behavior and force distribution for streaming spray transfer and projected spray transfer, separately. The arc area in streaming spray transfer mode with low CO₂ content (5% or 10%) is larger than that in projected spray transfer mode with high CO₂ content (20% or 25%), which provides the wire tip with more heat. The wire tip is sharper in streaming spray than in projected spray. Considering the similar gas property, the surface tension of the liquid metal is similar and the difference of F_γ is not distinguishable. According to Eqs. (4) and (5), F_{env} in Figure 8(b) is larger than that in Figure 8(a). However,

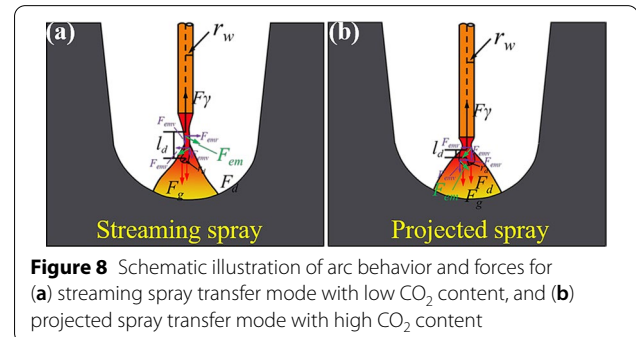


Figure 8 Schematic illustration of arc behavior and forces for (a) streaming spray transfer mode with low CO₂ content, and (b) projected spray transfer mode with high CO₂ content

due to the sharper wire tip, the radial component of the electromagnetic force F_{emr} acting on the droplet in Figure 8(a) is huge, to cause a droplet to subject a strong pinch effect, so the droplet is not able to grow to a massive volume.

The metal transfer under different O_2 content is shown in Figure 9. For all shielding gas mixtures, the metal is transferred in pulse streaming spray transfer mode. When O_2 increases from 2% to 5%, a long neck and a taper-shaped wire tip are found in both cases. The long neck is broken at the junction of the neck and the wire tip, and subsequently a big droplet accompanied by some small droplets transfers into the welding pool. And the arc jumping phenomenon accompanies the streaming spray transfer mode. Spatters are seen in Figure 9(b) and (c). The explosive droplet transfer is observed in Figure 10(c) (at $t+2.67$ ms). This may be ascribed to the received heat in droplet and the pinch effect of large radial electro-magnetic force. The O_2 content increases the energy for arc ignition and the heat loss because of its high thermal conductivity, which brings about a greater constriction for the arc. Therefore, the current density becomes higher in arc root, and so does the temperature in arc center. On the other hand, O_2 is a kind of active gas that is easy to react with the metal elements in wire. Lots of heat is released by the exothermic reactions. The droplet can be heated to a higher temperature to cause an explosive droplet transfer.

3.2 Welding Process Stability

Figure 10 shows the welding current and voltage waveforms for varied contents of CO_2 in the shielding gas. When the CO_2 content is 5%, the welding current and arc voltage waveforms are not stable at the initial stage of the welding process. Some extremely-high peaks appear at this stage as shown in Figure 10(a). As the CO_2 content increases to 10% or more, only few high peaks appear, which illustrates that the welding process becomes more stable. The high peaks in welding current waveforms represent the occurrence of short-circuit. The valley values (less than 10 V) in arc voltage waveforms are caused by the decreased arc length when the droplet is transferred to the welding pool. As shown in Figure 10, with increasing CO_2 content, the valley values obviously increase, even when the welding process is very stable.

Figure 11 displays the welding current and voltage waveforms for varied O_2 contents in the shielding gas. The effect of O_2 on stabilizing the welding process is more obvious due to its stronger oxidizing property. When about 2% O_2 content is added into the shielding gas, the welding process is similar to that shown in Figure 10(a). With O_2 content further increasing to 5% and 10%, the high peaks no longer appear as shown in Figure 11(b) and (c). It is worthwhile to note that the welding process is more effectively stabilized by the same content of O_2 than CO_2 .

Figure 12(a) displays the root-mean-square (RMS) current and voltage of welding in shielding gas with varied CO_2 contents. When the CO_2 content is in the range

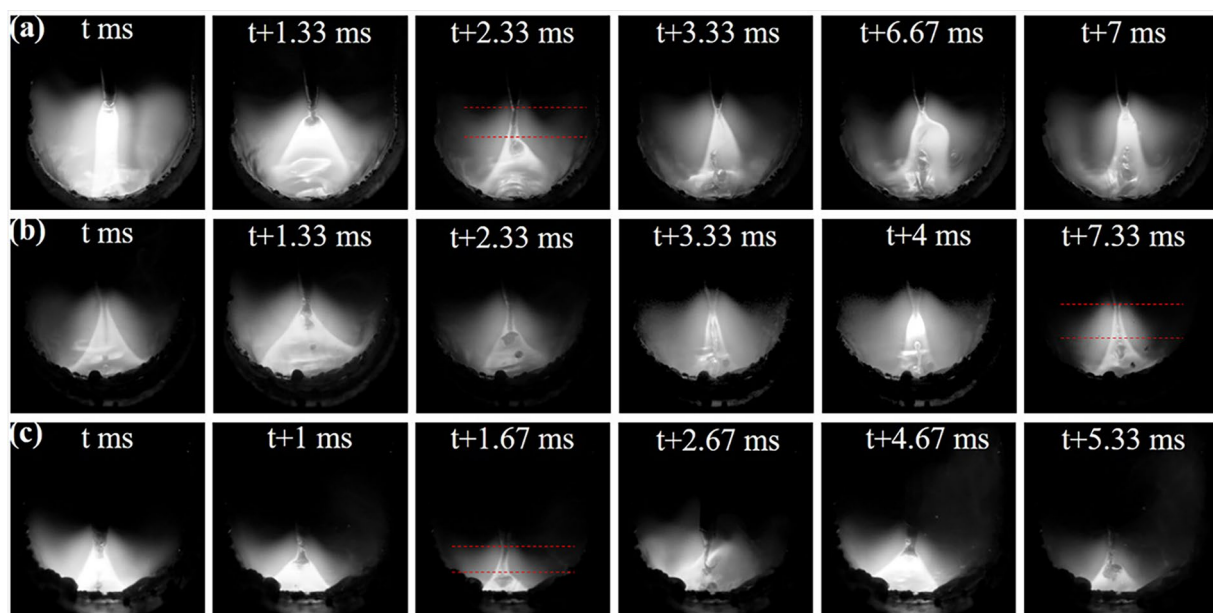


Figure 9 Metal transfer process under shielding of gas mixtures of (a) Ar + 2% O_2 , (b) Ar + 5% O_2 , (c) Ar + 10% O_2

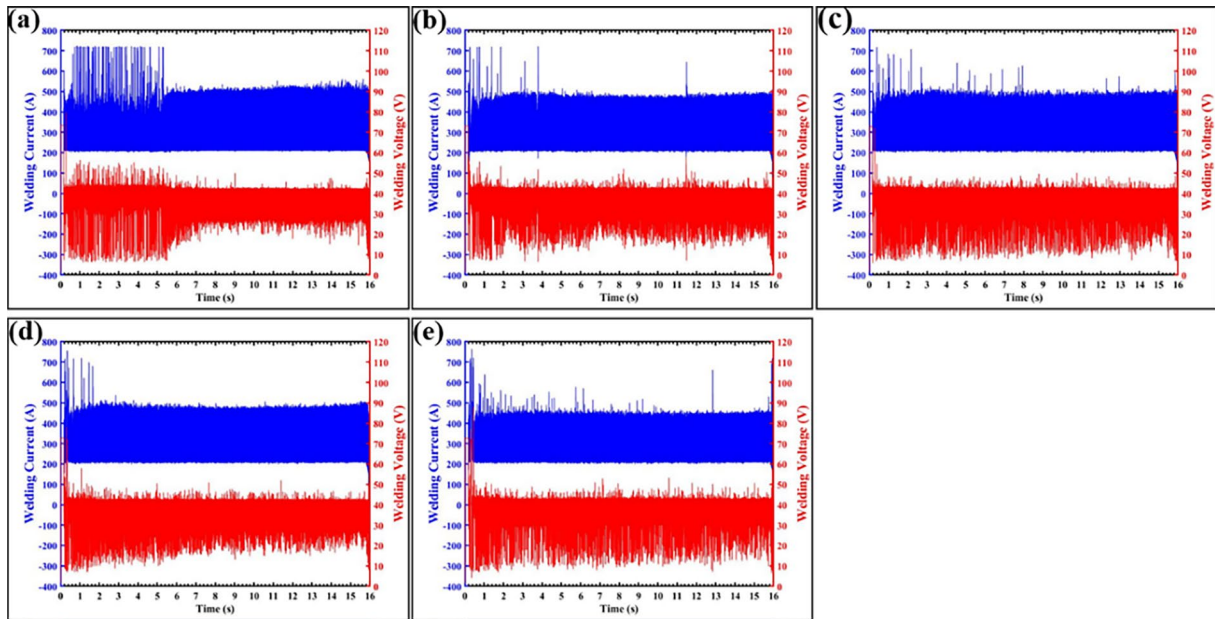


Figure 10 Current and voltage waveforms under shielding of (a) Ar + 5% CO₂; (b) Ar + 10% CO₂; (c) Ar + 15% CO₂; (d) Ar + 20% CO₂; (e) Ar + 25% CO₂

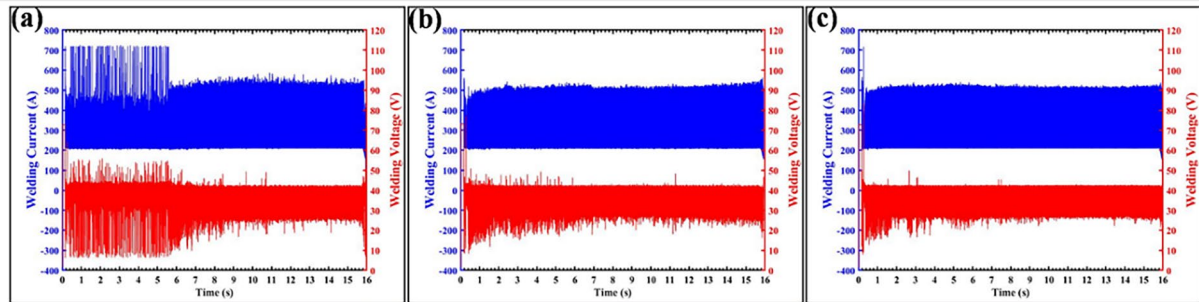


Figure 11 Current and voltage waveforms under shielding of (a) Ar + 2% O₂; (b) Ar + 5% O₂; (c) Ar + 10% O₂

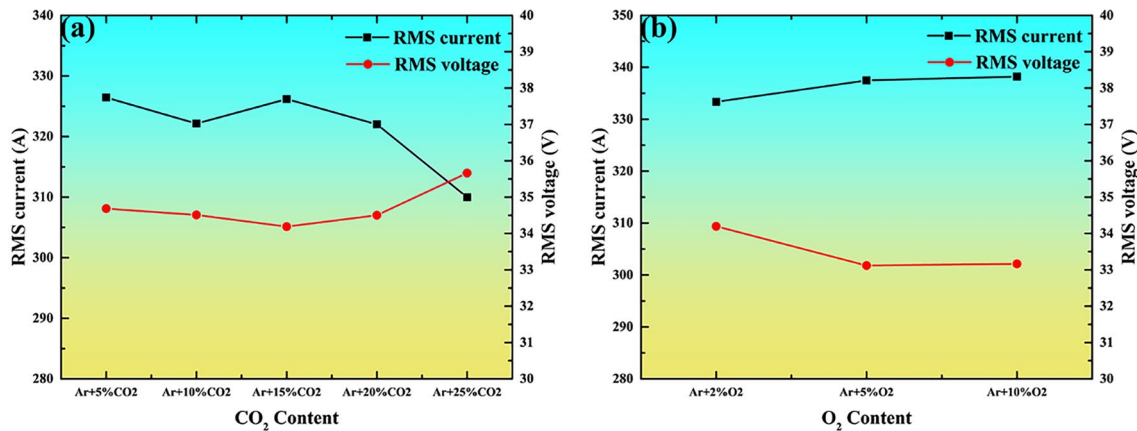


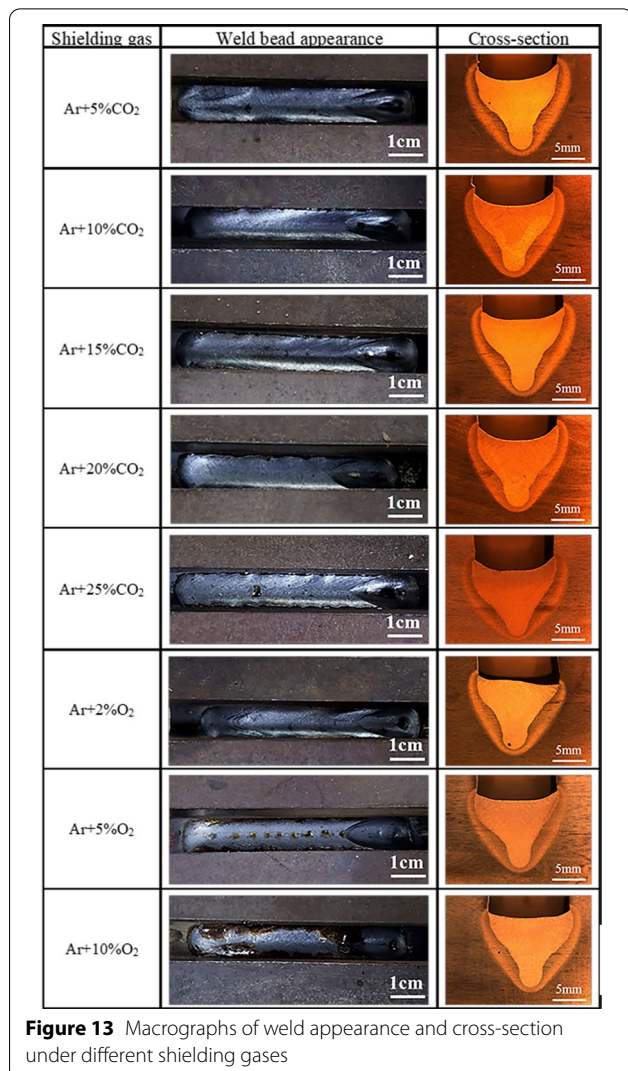
Figure 12 Variations of RMS current and voltage under shielding of gas with (a) varied CO₂ contents and (b) varied O₂ contents

from 5% to 25%, the RMS current shows a slight fluctuation first and then decreases while the RMS voltage first declines mildly and then takes on an obvious rising trend.

When the CO_2 content varies from 5% to 15%, the dissociation of CO_2 needs more energy, and meanwhile, the thermal conductivity of shielding gas also increases, which means more conductive energy loss. The increased total energy loss leads to constricting arc profile to make an energy balance. Therefore, the electron density of arc plasma has no significant changes, and so does the RMS current. The variation of arc length improves its potential gradient to maintain the arc stability. Therefore, the RMS voltage changes a little. With the CO_2 content further increasing to 25%, the energy loss continues to increase, and thus the ionization degree of shielding gas decreases, which results in the decrease of RMS current. To maintain a stable arc combustion, the RMS voltage increases eventually. Figure 12(b) shows the RMS current and voltage of welding in shielding gas with varied O_2 contents. The RMS current slightly increases while the RMS voltage decreases with the increase of the O_2 contents. O_2 has stronger oxidizability. Lots of heat is released by the exothermic reactions between O_2 and the metal elements of wire, as mentioned above. When the O_2 content is low, the ionization degree of shielding gas increases with increasing O_2 content, leading to increasing RMS current. In addition, the RMS current in shielding gas with certain O_2 content is larger than that in shielding gas with same CO_2 content. It is proved that the decrease of CO_2 could lower the transition current of projected spray transfer or streaming spray transfer. In other words, it is easier to obtain streaming spray transfer or projected spray transfer in the shielding gas with low CO_2 content.

3.3 Weld Bead Formation

Figure 13 shows the first weld bead appearances and their cross-sections of NG-GMAW under different shielding gases. When the CO_2 content is 5%, the weld bead surface is fluctuant, which may result from the random fluctuation of welding pool due to the unstable pulse streaming spray mode at the initial stage. When the CO_2 content increases to more than 5%, the weld bead appearance becomes smooth. When the O_2 content in the shielding gas gets up to 2%, the weld bead appearance is similar to that for the shielding gas of $\text{Ar} + 5\% \text{CO}_2$. In this case, the arc is long and its arc stiffness is relatively low. Hence, a droplet may not directly fall into the welding pool along the axis direction, as it is prone to be influenced by the anode spot force and other forces. When the O_2 content is 5%, some slags appear on the surface of the weld, distributing mainly along the centerline. With the O_2 content increasing to 10%, there are more slags on the weld surface. It is very difficult to remove the slags in



NG-GMAW. Even a few slags on the weld bead surface may cause inclusion defects or lack-of-fusion defects in subsequent multi-pass welding process, which will seriously damage the mechanical properties of the welding joint. Therefore, the shielding gas mixture with O_2 content over 5% is not suggested, although it can greatly improve the arc stability and the welding process.

The finger penetrations are obtained for all the shielding gas mixtures. The weld bead surface is nearly flat and an undercut defect occurs in $\text{Ar} + 2\% \text{O}_2$ shielding gas mixture, while the weld bead surfaces in other shielding gas mixtures appear to be concave. There is no lack-of-sidewall-fusion defect, and enough sidewall penetration is obtained for all weld beads. The weld width, the weld penetration and the weld surface concavity obtained in the shielding gases with varied CO_2 and O_2 contents are presented in Figure 14. Both the weld width and the weld penetration for all shielding gas mixtures are similar. The

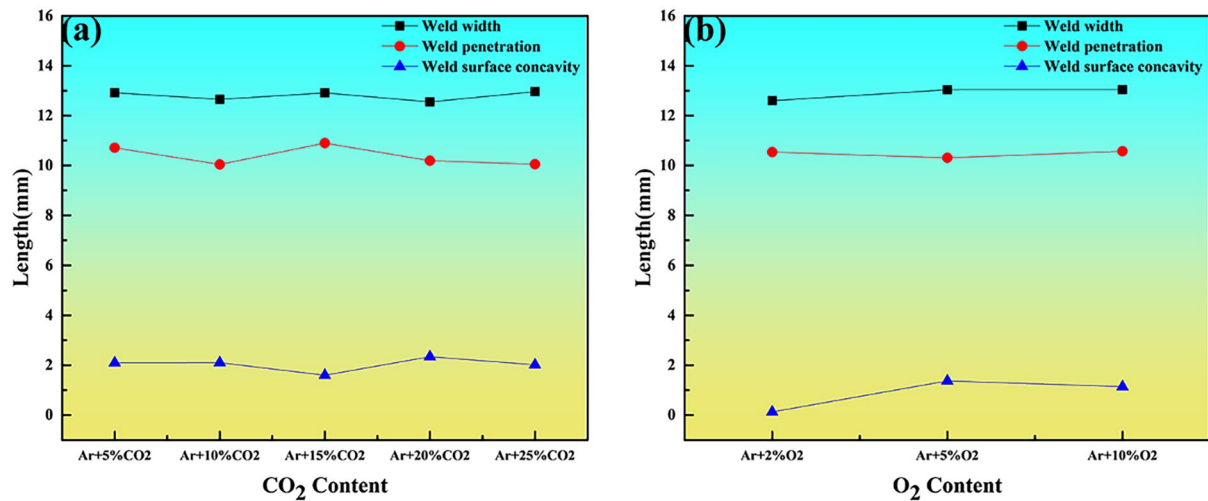


Figure 14 Variations of weld bead width, penetration and surface concavity with different (a) CO₂ content and (b) O₂ content

weld surface concavity can be evaluated based on the cross-section of the weld beads. It is in the range from 1.0 to 2.0 mm, which means a concave-shape weld bead surface.

The formation of the finger penetration and concave-shape weld bead surface is tightly related with the droplet transfer mode and arc pressure. The streaming spray transfer or spray transfer is the main transfer mode, in which the impact force is large enough to form a finger penetration, despite the relatively small droplet volume. According to Chen and Wu [30], the impact force of droplet is expressed as follow:

$$F_{imp} = \frac{f_d m_d V_d}{\pi R_d^2}, \quad (6)$$

where f_d is the droplet transfer frequency, m_d is the mass of the droplet, and V_d is the velocity of droplet impinging on the welding pool.

The flow behavior of the welding pool under different droplet transfer mode is shown in Figure 15. The droplet is smaller, and its transfer frequency is higher in streaming spray transfer mode. According to equation (6), the droplet impact force acting on the welding pool is very large, which results in a downward vortex and sunken welding pool surface given in Figure 15(a) and is easy to produce a finger penetration and concave-shape weld bead surface. As shown in Figure 15(b), due to the larger droplet and lower droplet transfer frequency, the droplet impacting force in this transfer mode is much less than that in streaming spray transfer mode. However, the arc length in projected spray transfer mode is smaller. Fan et al reported that the arc pressure increased with the decrease of arc length [31], which means larger arc

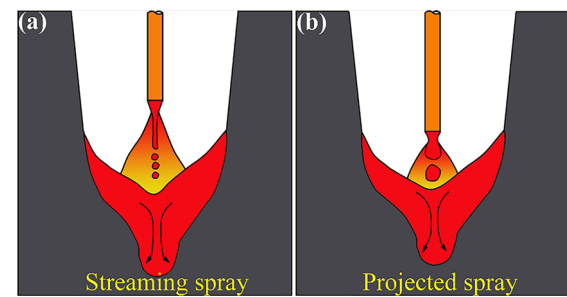


Figure 15 Flow behavior of the welding pool under (a) streaming spray droplet transfer and (b) projected spray droplet transfer

pressure in Figure 15(b), and thus the similar weld formation could form.

4 Conclusions

- (1) The arc length, width and area all decrease with the increase of CO₂ content or O₂ content. The thermal pinch effect and increased electromagnetic forces lead to a constricted arc when the CO₂ or O₂ content is higher. When the CO₂ content is in the range of 5%–10% or O₂ content is 5%, the arc edge is upward warped at the peak current due to the restrain effect of narrow gap groove.
- (2) With the CO₂ content in the shielding gas increasing from 5% to 25%, the droplet transfer mode changes from pulsed streaming spray to pulsed projected spray because of the increased heat loss and arc ignition energy. The droplet transfer mode keeps streaming spray transfer when the O₂ content ranges from 2% to 10%. The streaming spray trans-

fer is accompanied by the long neck and arc jumping phenomenon.

- (3) The welding process stability is improved first and then gets worse with the CO₂ content increasing from 5% to 25% while it becomes increasingly better and better with the O₂ content ranging from 2% to 10%. The effect of O₂ has a more obvious effect on stabilizing the welding process due to its stronger oxidizing property.
- (4) A concave weld bead surface is obtained in all shielding gases except Ar + 2% O₂. The weld bead surface is fluctuant when the CO₂ content is 5% or O₂ content is 2% resulting from the unstable metal transfer and arc behaviors.

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Authors' contributions

XT developed the idea, analyzed data, co-wrote and revised the paper. GL developed the idea, designed and performed the experiments, analyzed data and wrote the paper. QX performed the experiments. HC assisted with the experiments. FL assisted with paper revision. All authors read and approved the final manuscript.

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Competing interests

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